

Atom-by-Atom Analysis and Lithography in the Electron Microscope

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A modern, aberration-corrected Scanning Transmission Electron Microscope (STEM) equipped with an efficient Cold Field Emission Gun (CFEG) such as the Nion UltraSTEM [1] can form a probe of 60-200 keV electrons that is $\sim 1 \text{ \AA}$ in diameter and contains $\sim 160 \text{ pA}$ of current. This means that a single atom in a sample can be “illuminated” by $\sim 10^9$ medium-energy electrons per second, and that many different analytical signals can be collected from the single atom. Optimized post-sample optics and detectors allow the various signals to be captured efficiently, often with single particle sensitivity and close to 100% collection efficiency. The signals include forward scattered (bright-field) electrons as well as medium-angle and High-Angle Dark Field (HADF) electrons, which carry information about the types of atoms present in the sample and their locations; energy-loss electrons, which carry information about electronic states of the sample, the types of atoms present and their bonding arrangements, and also the sample’s vibrational (phonon) energies; and further emitted X-rays, which carry simple-to-interpret information about the types of atoms present in the sample. The cross-sections for some of these signals are rather large, allowing the signals to be collected in 10^{-6} to 10^{-3} seconds per examined atom.

When the energy of the primary electrons is below the knock-on threshold in materials that do not suffer ionization damage, there is essentially no radiation damage and data with very high signal-to-noise ratios becomes available. In graphene and monolayer BN this means operating at $\sim 60 \text{ keV}$ or lower primary energy. By raising the primary energy to $\sim 100 \text{ keV}$, it becomes possible to knock out individual atoms from the structure, thereby creating single vacancies and more complicated defects. Some of the defects subsequently trap atoms diffusing on the surface [2, 3]. If the diffusing atoms are introduced deliberately, it becomes possible to select the dopant atom for a specific site.

Figure 1 shows a 60 keV HADF image of graphene containing substitutional Si atoms in two different configurations: a one-Si-for-one-C substitution with 3-fold coordination, and a one-Si-for-two-Cs substitution with 4-fold coordination. Figure 2 shows Electron Energy Loss Spectra (EELS) collected from the two types of silicon atoms with a dose of about $\sim 10^{10}$ electrons. The dose was large enough to capture EELS fine structures that allowed the detailed electronic state of the single Si atoms to be determined. This showed that the 3-coordinated Si atom sticks out of the graphene plane [4, 5].

Planned modification of the Nion UltraSTEM will allow different atomic species to be introduced in-situ while the sample is being observed, thus enabling defects such as those shown in Fig. 1 to be created and patterned into new types of structures and devices.

[1] N. Dellby et al., *Eur. Phys. J. Appl. Phys.* **54** (2011) 33505.

[2] O.L. Krivanek et al., *Nature* **464** (2010) 571–574 (see supporting materials).

[3] R. Zan, Q.M. Ramasse, U. Bangert and K.S. Novoselov, *Nano Lett.* **12** (2012) 3936–3940.

[4] W. Zhou et al., *Phys. Rev. Lett.* **109** (2012) 206803.

[5] Q.M. Ramasse et al., *Nano Lett.* **13** (2013) 4989–4995.

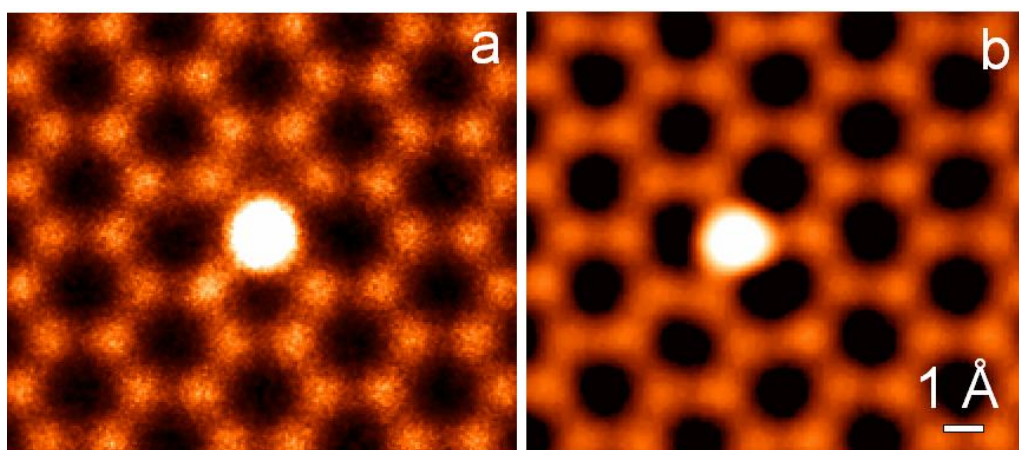


Figure 1. 60 keV HADF STEM images of a) Si substituting for two C atoms in monolayer graphene, b) Si substituting for one C atom.

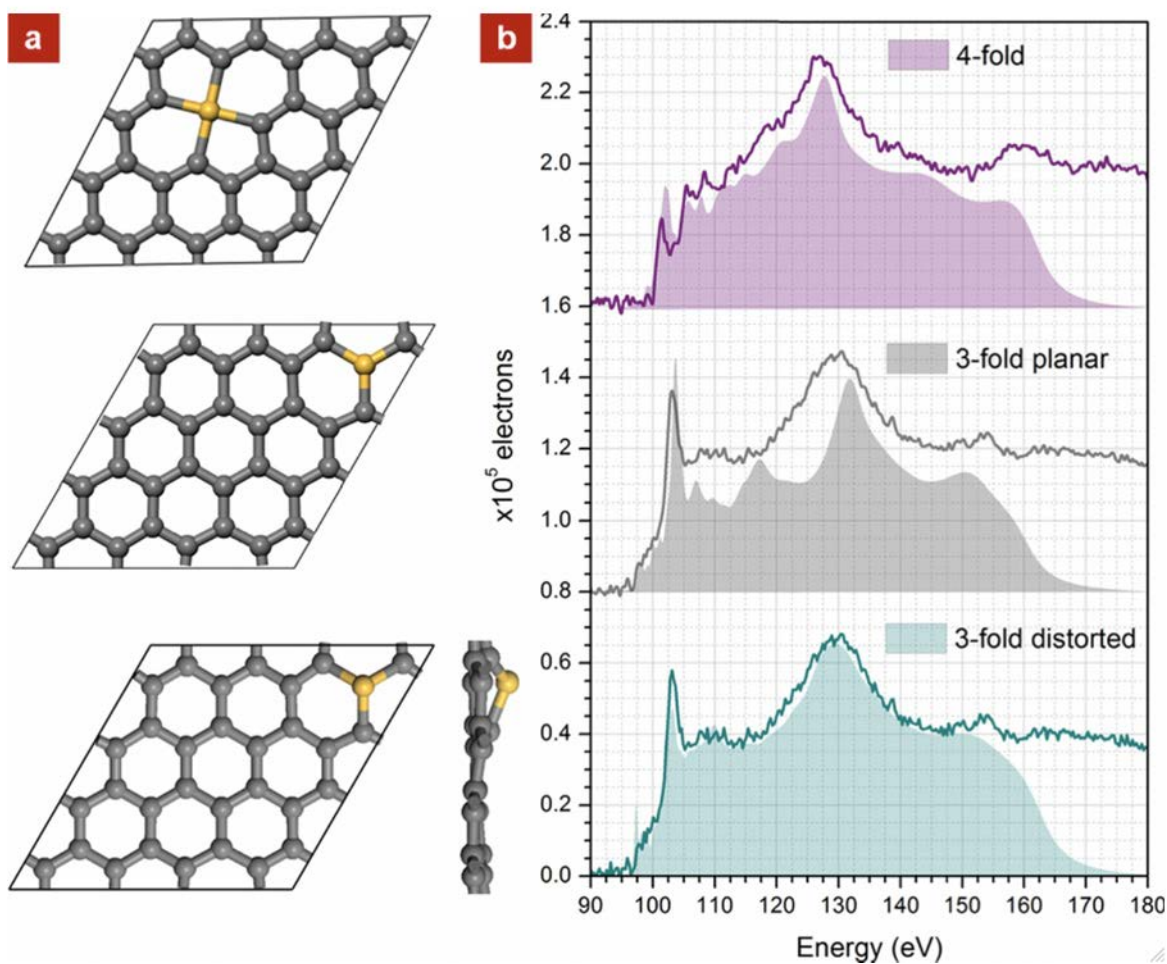


Figure 2. EELS data and atomic models for the two types of Si substitutional atoms shown in Fig. 1. a) Atomic models. b) Fine structure of Si $L_{2,3}$ edge EEL spectra from individual Si atoms (lines) and CASTEP simulations for the models shown in (a) (solid spectra).